Mean transverse energy of ultrananocrystalline diamond photocathode

Cite as: Appl. Phys. Lett. **114**, 093103 (2019); https://doi.org/10.1063/1.5084167 Submitted: 04 December 2018 . Accepted: 16 February 2019 . Published Online: 05 March 2019

Gongxiaohui Chen, Gowri Adhikari, Linda Spentzouris, Kiran Kumar Kovi ៉, Sergey Antipov, Chunguang Jing, W. Andreas Schroeder, and Sergey V. Baryshev



ARTICLES YOU MAY BE INTERESTED IN

Electronic structure of SnSe₂ films grown by molecular beam epitaxy Applied Physics Letters **114**, 091602 (2019); https://doi.org/10.1063/1.5084147

Wafer-scale on-chip synthesis and field emission properties of vertically aligned boron nitride based nanofiber arrays

Applied Physics Letters 114, 093101 (2019); https://doi.org/10.1063/1.5079655

Workfunction fluctuations in polycrystalline TiN observed with KPFM and their impact on MOSFETs variability Applied Physics Letters 114, 093502 (2019); https://doi.org/10.1063/1.5090855





Lake Shore

8600 Series VSM For fast, highly sensitive measurement performance LEARN MORE •



Appl. Phys. Lett. **114**, 093103 (2019); https://doi.org/10.1063/1.5084167 © 2019 Author(s). Cite as: Appl. Phys. Lett. **114**, 093103 (2019); doi: 10.1063/1.5084167 Submitted: 4 December 2018 · Accepted: 16 February 2019 · Published Online: 5 March 2019



Gongxiaohui Chen,^{1,a)} Gowri Adhikari,² Linda Spentzouris,¹ Kiran Kumar Kovi,³ (b) Sergey Antipov,³ Chunguang Jing,³ W. Andreas Schroeder,^{2,b)} and Sergey V. Baryshev^{4,c)}

AFFILIATIONS

¹Department of Physics, Illinois Institute of Technology, Chicago, Illinois 60616, USA

- ²Department of Physics, University of Illinois at Chicago, Chicago, Illinois 60607, USA
- ³Euclid TechLabs LLC, Bolingbrook, Illinois 60440, USA

⁴Department of Electrical and Computer Engineering, Michigan State University, East Lansing, Michigan 48824, USA

- ^{a)}Electronic mail: gchen26@hawk.iit.edu
- ^{b)}Electronic mail: andreas@uic.edu
- ^{c)}Electronic mail: serbar@msu.edu

ABSTRACT

Nitrogen incorporated ultrananocrystalline diamond [(N)UNCD] could be an enabling material platform for injector photocathode applications due to its high emissivity. While the quantum efficiency (QE) of UNCD was reported by many groups, no experimental measurements of the intrinsic emittance/mean transverse energy (MTE) have been reported. Here, MTE measurement results for an (N)UNCD photocathode in the photon energy range from 4.41 to 5.26 eV are described. The MTE demonstrates no noticeable dependence on the photon energy, with an average value of 266 meV. This spectral behavior is shown not to be dependent upon physical or chemical surface roughness and inconsistent with low electron effective mass emission from graphitic grain boundaries but may be associated with emission from spatially confined states in the grain boundaries. The combined effect of low and constant MTE and high QE, which increases with respect to the excess laser energy, may pave the way for bright UNCD photocathodes for electron injectors.

Published under license by AIP Publishing. https://doi.org/10.1063/1.5084167

Photocathode-based RF and pulsed DC guns are bright electron injectors for free electron lasers and advanced time resolved microscopes.¹ Further progress of electron laser and microscopy facilities (improved sensitivity, spatiotemporal resolution, and high throughput) largely depends on the development and understanding of materials with the potential to be utilized as photocathodes. Photocathode development challenges include achieving simultaneously (*i*) high quantum efficiency (QE), (*ii*) high transverse coherence [meaning low intrinsic emittance/low mean transverse energy (MTE)], and (*iii*) rapid response time.

The ratio of the charge to the MTE determines the photocathode brightness, which in many applications is the most critical figure of merit. For a classical metal photocathode such as copper, the Fowler-Dubridge law² predicts that the emitted charge is a fast-growing function of excess energy (a power law), where excess energy ΔE is the difference between the laser primary incident photon energy $\hbar\omega$ and the work function ϕ defined as $\Delta E = \hbar\omega - \phi$. Dowell and Schmerge³ have found that the transverse momentum for metals also grows with

excess energy as $\sim \sqrt{\hbar \omega - \phi}$. For the latter reason, to attain the highest quality (low divergence) electron beam metal photocathodes are often operated in the near threshold region (having the smallest ΔE , with the primary photon energy nearly matching the work function), although brightness increases with excess energy.

A great number of metal and thin film alkali antimonide photocathodes obey the Dowell-Schmerge (DS) model.^{3–5} However, some semiconductor photocathodes, e.g., GaAs and PbTe, show various MTE versus excess energy trends that are different from those specific to metals. Negative electron affinity (NEA) GaAs photocathodes,⁶ for instance, demonstrate ~1000-fold QE increase as the excess energy increases from 0 to about 1 eV while the MTE remains low and nearly constant with the same ΔE range (within measurement precision).

Nitrogen incorporated ultrananocrystalline diamond [(N)UNCD] is another example of a NEA photocathode that has high electron conductivity through the bulk of a semi-metallic nature. The NEA is induced via surface C-H dipole formation when UNCD is processed in a hydrogen plasma (UNCD:H).^{7,8} Compared to cesiated GaAs, the

ARTICLE

C–H dipole is stable in air. The resulting QE is high ($\sim 10^{-3}$)^{9–11} and could potentially be further increased by reducing the work function through using a different *n*-type dopant such as phosphorous.¹² The intrinsic as-grown surface roughness is low, less than 10 nm. The low physical roughness suggests that the beam emittance can be low. In order to elucidate the ultimate performance of *n*-type UNCD, the transverse electron momentum Δp_T is a fundamental parameter to determine (or equivalently the MTE, since MTE = $\Delta p_T^2/2m$, with *m* being the electron mass).

In this paper, the MTE of a (N)UNCD photocathode was experimentally measured over an excess energy range of 1 eV. It was found that the MTE does not noticeably depend on the excess energy with an average value of 266 meV. It is proposed that this spectral behavior is due to emission from spatially confined states in the graphite regions (i.e., grain boundaries) between the diamond grains.

The (N)UNCD film, 160 nm thick, was synthesized by a microwave plasma chemical vapor deposition method. The film was deposited onto highly doped n-Si substrate to avoid charging effects during photoemission measurements. Deposition parameters were identical to those reported in the paper of Perez Quintero et al.⁹ For the injector photocathode operated in UV/visible range, a (N)UNCD film thickness of 150-200 nm is optimal in terms of absorbing light and generating photoelectrons because for wavelengths >100 nm light is absorbed at the length scale of the photoelectron escape depth which is approximately 50 nm.¹³ The resistivity of the film was assumed to be 0.1 Ω \times cm, as suggested by a four-probe measurement of a (N)UNCD film grown on an insulating Si witness coupon under the same growth conditions. A Raman spectrum confirming the (N)UNCD chemical bonding structure is presented in Fig. 1(a). The spectrum suggests that the amount of graphitic grain boundary sp^2 -phase in the sp^3 diamond matrix is approximately 4%.¹⁴ The work function of (N)UNCD film was found to be 4.2 eV using a Kelvin probe.

The experimental setup for measuring the MTE of the emitted electrons using the solenoid scan technique¹⁵ is shown in Fig. 1(b).

The electron pulses are generated in a 20 kV DC gun using tunable ultraviolet (UV) radiation from a 30 MHz repetition rate subpicosecond laser system driven by a diode pumped and mode-locked Yb:KGW oscillator.¹⁶ Briefly, the 1046 nm, 0.25 ps pulse duration output from the 2 W Yb:KGW oscillator is used to generate a continuum in a photonic crystal fiber which is then selectively amplified by optical parametric amplification (OPA) in lithium triborate (LBO) nonlinear crystals and the resulting signal and idler pulses are subsequently sum frequency mixed with the second and third harmonics of the Yb:KGW laser to generate tunable UV radiation. Together with the 3.56 eV (349 nm) third and 4.75 eV (262 nm) fourth harmonics of the Yb:KGW laser, this provides a UV radiation source with almost continuous 3.0-5.3 eV (235-410 nm) tunability. The p-polarized UV laser beam is circular for the third and fourth harmonics and elliptical (aspect ratio ~1:1.4) from the sum frequency generation. At the employed 60° incidence angle, the measured half-width 1/e maximum, dubbed as $HWe^{-1}M$, (x, y) irradiance spot sizes on the photocathode surface are (220, 80) \pm 5 μ m for the sum frequency generated UV radiation and (240, 120) \pm 5 μ m for the fourth harmonic at 4.75 eV. The known electron source size then provides a required input parameter for the extended analytical Gaussian (AG) simulation^{17,18} of the electron beam propagation from emission to detection, with a Ce:GAGG scintillation crystal and a CCD camera, through the solenoid scan MTE measurement system. Figure 2 displays the measured $HWe^{-1}M$ electron beam sizes at the scintillator as a function of the square of the current (i.e., focal strength) passing through the two solenoid lenses (counter wound to avoid image rotation effects) for the 4.75 eV incident UV photon energy. In this case, the AG model simulation fit using either the horizontal (x) or vertical (y) spot size (range

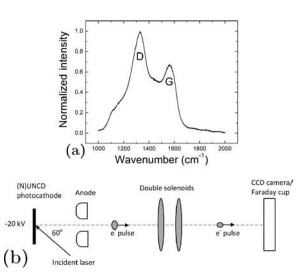


FIG. 1. (a) Raman spectrum of the (N)UNCD sample showing the characteristic diamond (D) and graphite (G) peaks; (b) The solenoid scan system for MTE measurement. Electrons accelerated in the DC gun travel through the double-solenoid lens and imping on the scintillator screen imaged by a CCD camera.

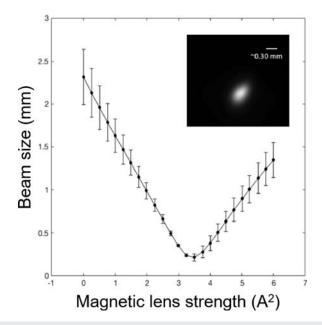


FIG. 2. Solenoid scan data for an incident 4.75 eV photon energy. The top and bottom of the "error bars" represent the measured horizontal and vertical $HWe^{-1}M$ beam sizes, respectively, and the connected dots the average beam size. The inset shows the actual beam image at the focal point for the solenoid scan, i.e., at a solenoid current of 1.83 A.

indicated by the "error bar span" in Fig. 2), or the average beam size (connected dots in Fig. 2), generates an MTE value for the emitted electrons of 290(\pm 40) meV. We note that the measurements were made in the low charge regime so that the beam imaging and MTE calculations were not affected by space charge. The low charge regime refers to less than 10 electrons per pulse. Our electron pulse propagation simulation of the experimental system indicates that space-charge effects only become significant for more than 1000 electrons per pulse. Additionally, the low 0.45 MV/m acceleration field gradient on the (N)UNCD photocathode surface in the DC gun ensured that dark current effects were negligible and produced a Schottky effect of only \sim 30 meV that is comparable to the thermal energy at room temperature.

Figure 3 presents the full summary of MTE values obtained through multiple solenoid scans performed at multiple primary laser photon energies. The measured MTE values display a near flat trend, remaining independent of the UV photon energy in the range of 4.41 to 5.26 eV. The red dashed line in Fig. 3 represents the mean MTE value of 266 meV. Such a flat response is not common for most photocathode systems. To compare the deduced MTE($\hbar\omega$) dependence of (N)UNCD against the DS model, ($\hbar\omega - \phi$)/3 (blue solid line) is plotted using $\phi = 4.2 \text{ eV}$. The drastic deviation from the metallic photoemission model calls for an alternative scenario.

As per Raman spectrum in Fig. 1(a), (N)UNCD is a graphitic rich two-phase material that consists of sp^3 diamond grains and sp^2 graphitic grain boundaries. It has been reported that electron emission (photo- or field-emission) in this and related materials preferentially originates from the grain boundaries.^{19,20} Accordingly, a possible explanation for the observed spectral dependence of the MTE is physical and chemical roughness associated with the nano-granularity of the photocathode material. However, the analysis presented in the paper of Karkare and Bazarov²¹ suggests that neither play a role in our case. The low ~ 0.5 MV/m surface acceleration field in the 20 kV DC gun excludes any significant increase in MTE due to the 10 nm surface roughness associated with the UNCD grain size. Similarly, effects on the MTE due to chemical roughness are negligible despite the difference in diamond sp^3 and graphitic sp^2 grain boundary work functions, which can be as large as 1.5 eV. The latter is due to the rapid decay from the photocathode surface of transverse field modulations due to

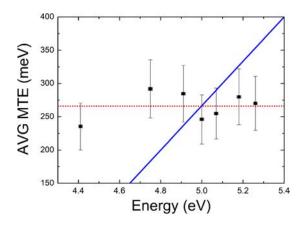


FIG. 3. MTE values extracted from the solenoid scan data plotted versus the incident photon energy. The red dotted line depicts the mean MTE equal to 266 meV. The blue line is the DS model for which $(\hbar \omega - \phi)/3$.

work function variations for the small ${\sim}10\,\mathrm{nm}$ scale (grain size) periodicity.

Studies of the bulk electron transport in (N)UNCD have confirmed that electrons percolate through grain boundary networks with an effective mass equivalent to that in graphite,²² i.e., an electron effective mass 1/18 of the electron rest mass m_0 . Such a small electron mass in turn suggests narrow electronic energy bands, which could affect the MTE of electron emission. Consider a one-step quantum mechanical photoemission mechanism where the incident photon momentum is considered negligible, so that the excited virtual states (intermediate states between the electron states inside the material and the free space electron states) have the same dispersion relation $\varepsilon(k)$ as in the initial state, i.e., as inside the material before photon absorption. Figure 4 compares such a one-step photoemission process from states with a dispersion determined by the rest electron mass m_0 (a) to one from states with a small effective mass m^* (b). With transverse momentum conservation in photoemission, it is clear that a small effective mass for the emitting states can serve to restrict the transverse momentum, and hence the MTE, of the photo-emitted electrons [Fig. 4(b)]. In contrast, the transverse momentum for an electron emitted from a "perfect" metal photocathode with a free electron mass dispersion is only restricted by the vacuum state dispersion [Fig. 4(a)], resulting in a

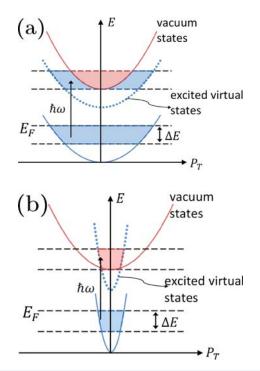


FIG. 4. Simplified one-step photoemission (energy versus transverse momentum) diagrams that capture the effective mass effect on MTE: (a) photoemission from states with dispersion corresponding to free electron mass m_0 resulting in DS like behavior and (b) photoemission from states with a small effective electron mass m^* resulting in MTE behavior insensitive to the excess energy. The solid blue lines represent the actual metallic-like parabolic $\varepsilon(k)$ dispersion relations, where the blue highlighted area indicates states with sufficient excess energy for photoemission. The dotted blue line represents the virtual excited states, and the red highlighted area depicts the final allowed photo-emitting electron states. Adapted with permission from Ref. 26.

MTE($\hbar\omega$) variation in accordance with the DS model. A rule-ofthumb proposed in the paper of Rickman et al.²³ is that transverse electron momentum p_T in the photoemission process that is actually realized in experiment is restricted by either $\sqrt{2m_0(\hbar\omega-\phi)}$ or $\sqrt{2m^*E_F}$ (where E_F is the Fermi energy), whichever is smaller. Since for graphite $m^* = 0.045 m_0^{24}$ and E_F may be as small as 30 meV,²⁵ the product m^*E_F is always less than the product $m_0\Delta E$. From this simplified consideration, the MTE should be dependent upon the Fermi energy, but the measured quantity is almost an order of magnitude larger.

An alternative explanation for the measured relatively invariant MTE of above threshold electron emission (Fig. 3), and consistent with the above, is that the emitted electrons originate from spatially confined states in the graphite regions between the diamond grains. If so, transverse momentum conservation in photoemission implies that the MTE should reflect the rms momentum (i.e., size in momentum space) of these states, assuming surface effects can be neglected. To obtain the observed MTE of \sim 270 meV, the emitting states would then need to be confined to a spatial region of ~1 nm-this is obtained through the Heisenberg's uncertainty principle. The emitter size of \sim 1 nm is well comparable with the grain boundary size in (N)UNCD, because in nitrogen incorporated UNCD average grain size is 10 nm and grain-boundary width is 1 nm.²⁷ Further experimental and theoretical investigations will be needed to shed light on the exact photoemission mechanism and hence the realized MTE values: for example, atomic resolution electron energy loss spectroscopy of the graphite grain boundaries and detailed band structure studies of (N)UNCD.

In summary, we reported the measurement results of MTE for a (N)UNCD photocathode. No noticeable dependence of the MTE on the excess energy over a range of 1 eV was measured, which is nonconventional behavior that has been observed so far in only a few photocathode systems. This spectral dependence is shown not to be dependent upon surface roughness (physical or chemical) and inconsistent with low electron effective mass emission from graphitic grain boundaries, but is likely to be associated with emission from spatially confined states in the \sim 1 nm graphite regions between the diamond grains. Given this promising intrinsic emission property of (N)UNCD, the next step would be to measure the MTE and QE of a hydrogen surface terminated (N)UNCD sample. If the MTE remains nearly constant, while the QE increases as expected due to the NEA produced after the hydrogen termination, then the (N)UNCD material has great potential as a next generation photocathode, given its additional ability to withstand poor vacuum without significant degradation.

This project was supported by the NSF Grant No. NSF-1739150, DOE SBIR program Grant No. DE-SC0013145 and NSF Grant No. PHYS-1535279. The use of the Center for Nanoscale Materials, an Office of Science user facility, was supported by the U.S. Department of Energy, Office of Science, Office of Basic Energy Sciences, under

Contract No. DE-AC02-06CH11357. S.V.B. was supported by funding from the College of Engineering, Michigan State University, under the Global Impact Initiative.

REFERENCES

¹D. H. Dowell, I. Bazarov, B. Dunham, K. Harkay, C. Hernandez-Garcia, R. Legg, H. Padmore, T. Rao, J. Smedley, and W. Wan, Nucl. Instrum. Methods Phys. Res. Sect. A 622, 685 (2010).

- ²L. A. DuBridge, Phys. Rev. 43, 727 (1933).
- ³D. H. Dowell and J. F. Schmerge, Phys. Rev. Spec. Top. Accel. Beams 12, 074201 (2009).
- ⁴I. Bazarov, L. Cultrera, A. Bartnik, B. Dunham, S. Karkare, Y. Li, X. Liu, J. Maxson, and W. Roussel, Appl. Phys. Lett. 98, 224101 (2011).
- ⁵J. Feng, J. Nasiatka, W. Wan, T. Vecchione, and H. A. Padmore, Rev. Sci. Instrum. 86, 015103 (2015).
- ⁶S. Karkare, L. Boulet, L. Cultrera, B. Dunham, X. Liu, W. Schaff, and I. Bazarov, Phys. Rev. Lett. 112, 097601 (2014).
- ⁷J. B. Cui, J. Ristein, and L. Ley, Phys. Rev. Lett. 81, 429 (1998).
- ⁸T. Sun, F. A. M. Koeck, C. Zhu, and R. J. Nemanich, Appl. Phys. Lett. 99, 202101 (2011).
- ⁹K. J. Pérez Quintero, S. Antipov, A. V. Sumant, C. Jing, and S. V. Baryshev, Appl. Phys. Lett. 105, 123103 (2014).
- ¹⁰S. Ohmagari, T. Hanada, Y. Katamune, S. Al-Riyami, and T. Yoshitake, Jpn. J. Appl. Phys., Part 1 53, 050307 (2014).
- ¹¹J.-P. Mazellier, C. Di Giola, P. Legagneux, C. Hébert, E. Scorsone, P. Bergonzo, and S. Saada, J. Vac. Sci. Technol. B 33, 03C105 (2015).
- ¹²F. A. Koeck, R. J. Nemanich, Y. Balasubramaniam, K. Haenen, and J. Sharp, Diamond Relat. Mater. 20, 1229 (2011).
- ¹³A. S. Tremsin and O. H. Siegmund, Diamond Relat. Mater. 14, 48 (2005).
- ¹⁴T. Corrigan, D. Gruen, A. Krauss, P. Zapol, and R. Chang, Diamond Relat. Mater. 11, 43 (2002).
- ¹⁵W. Graves, L. DiMauro, R. Heese, E. D. Johnson, J. Rose, J. Rudati, T. Shaftan, and B. Sheehy, "Measurement of thermal emittance for a copper photocathode," in Proceedings of the Particle Accelerator Conference (2001), Vol. 3, pp. 2227–2229.
- J. A. Berger, M. J. Greco, and W. A. Schroeder, Opt. Express 16, 8629 (2008).
- ¹⁷A. M. Michalik and J. E. Sipe, J. Appl. Phys. **99**, 054908 (2006).
- ¹⁸J. A. Berger and W. A. Schroeder, J. Appl. Phys. 108, 124905 (2010).
- ¹⁹R. L. Harniman, O. J. L. Fox, W. Janssen, S. Drijkoningen, K. Haenen, and P. W. May, Carbon 94, 386 (2015).
- ²⁰J. B. Cui, J. Ristein, and L. Ley, Phys. Rev. B **60**, 16135 (1999).
- ²¹S. Karkare and I. Bazarov, Phys. Rev. Appl. 4, 024015 (2015).
- ²²K. V. Shah, D. Churochkin, Z. Chiguvare, and S. Bhattacharyya, Phys. Rev. B 82, 184206 (2010).
- 23 B. L. Rickman, J. A. Berger, A. W. Nicholls, and W. A. Schroeder, Phys. Rev. Lett. 111, 237401 (2013).
- ²⁴R. C. Tatar and S. Rabii, Phys. Rev. B 25, 4126 (1982).
- ²⁵J. M. Schneider, "Electronic properties of graphite," Ph.D. thesis (Université Joseph-Fourier-Grenoble I, Grenoble, 2010).
- ²⁶W. A. Schroeder, T. Li, and B. Rickman, "PbTe(111): DFT analysis and experimental results," presented at the Photocathode Physics for Photoinjectors (P3) Workshop, Newort News, VA, 17-19 October 2016.
- ²⁷J. Birrell, J. A. Carlisle, O. Auciello, and D. Gruen, Appl. Phys. Lett. 81, 2235 (2002).